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## SSNP 160 – hydrogen Diffusion in a Summarized elastoplastic

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### steel:

This test of nonlinear quasi-static mechanics simulates the hydrogen diffusion in a homogeneous volume of steel according to plastic strains.

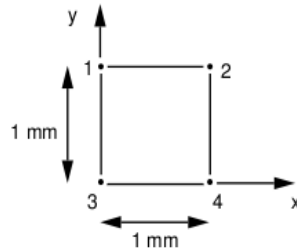
The modelization A is carried out with plane elements (mechanical modelization C\_PLAN, thermal modelization PLAN\_DIAG).

The modelization A is carried out with elements 3D (mechanical modelization 3D, thermal modelization 3D\_DIAG).

## 1 Problem of reference

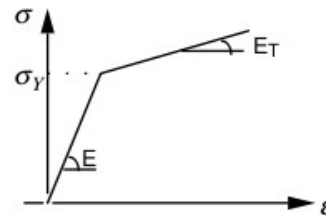
### 1.1 Geometry

the homogeneous solution is obtained on a plane volume element (square on side 1 ) or 3D (cubic on side 1 ). Dimensions are without influence on the solution.



### 1.2 Properties of the materials

$$\begin{aligned} E &= 2.10^7 \text{ Pa} \\ \nu &= 0.3 \\ \sigma_y &= 2.10^8 \text{ Pa} \\ E_T &= 2.10^9 \text{ Pa} \end{aligned}$$



For the diffusion, the coefficients material are, for a temperature of  $T = 293 \text{ K}$  :

$$D_L = 1.27 \cdot 10^{-8}$$

$$N_L = 5.1 \cdot 10^{29} \text{ m}^{-3}$$

$$K_T = e^{60000./RT}$$

$$a_1 = 23.26 \quad a_2 = -2.33 \quad a_3 = -5.5$$

$$V_h = 2. \cdot 10^{-6} \quad R = 8.3144 \text{ J/mol/K}$$

### 1.3 Boundary conditions and loadings

Side 1–3  $F_x$  imposed

Side 2–4 :  $u_x = 0$ .

Point: 2  $u_y = 0$ .

Loading by an increasing  $F_x$  distributed force according to time:

Time ( s )	0	$10^7$	$2 \cdot 10^7$
Force $F_x$	0	$\sigma_y$	$3 \sigma_y$

temporal Discretization:

- 1 step until  $t = 10^7$
- 100 steps until  $t = 2 \cdot 10^7$  : for each step, a mechanical computation then a computation of diffusion.

Initial concentrations in  $H_2$  :

- In the crystal lattice ("lattice")  $C_L(0) = 2.08 \cdot 10^{21} \text{ m}^{-3}$

- In the traps ("traps")  $C_T(0) = 8.42 \cdot 10^{20} \text{ m}^{-3}$

## 2 Reference solution

### 2.1 Method of calculating used for the reference solution

In the document [1], a model of diffusion of the hydrogen atoms in steels is suggested. He considers two types of concentration of the hydrogen atoms:

- $C_L$  is the concentration in the crystal lattice (Lattice)
- $C_T$  is the concentration in the "traps" or gaps (Traps)

Without taking again here all the approach followed by the authors of [1], the formulation suggested for the equation of diffusion of  $C_L$  is:

$$\frac{C_L + C_T(1 - \theta_T)}{C_L} \frac{\partial C_L}{\partial t} - \nabla \cdot (D_L \nabla C_L) + \nabla \cdot \left( \frac{D_L C_L V_H}{RT} \nabla \sigma_H \right) + \theta_T \frac{d N_T}{d \varepsilon_{eq}^p} \frac{d \varepsilon_{eq}^p}{dt} = 0 \quad \text{eq1}$$

One thus notes that the equation of diffusion takes account of the local gradient of the trace of the stresses (forced hydrostatic  $\sigma_H = 1/3 \text{tr}(\sigma)$ ) and of the equivalent plastic strain of Von Mises.

The relations defining the various quantities are:

$\theta_L = \frac{C_L}{N_L}$  is the occupancy rate of the sites of the crystal lattice, with  $N_T$  the number of sites per unit of volume.  $N_L$  is a constant estimated at  $N_L = 5,1 \cdot 10^{29} \text{ m}^{-3}$  for iron  $\alpha$  in [1].

$\theta_T = \frac{C_T}{N_T}$  the occupancy rate of the sites of the traps, with  $N_T$  density of the traps, i.e. the number of sites corresponding to traps per unit of volume.

Contrary to  $N_L$  which is a constant,  $N_T$  function of the plastic strain according to the statement:  $\log_{10}(N_T) = a_1 - a_2 \exp(-a_3 \varepsilon_{eq}^p)$ , with:  $a_1 = 23,26$ ,  $a_2 = 2,33$ ,  $a_3 = -5,5$  [1].

$D_L = 1,27 \cdot 10^{-8} \text{ m}^2/\text{s}$   $V_H = 210^{-6} \text{ m}^3$  for iron  $\alpha$  with room temperature,  $R = 8,3144 \text{ J/mol/K}$  is the constant of perfect gases, and  $T$  the temperature in  $^\circ \text{K}$ .

It remains to define  $C_T$  according to  $C_L$ : according to [1] with the equilibrium, which is the case for the CSC:

$$C_T = \frac{N_T}{1 + \frac{1}{K_T \theta_L}}, \text{ with } K_T = \exp(-\Delta E_T / RT) = 4,9710^{10} \text{ room temperature, following [1], while}$$

taking  $\Delta E_T = -60 \text{ KJ/mol}$ .

In a way similar to the nonlinear thermal, the variational formulation is written then:

Either  $\Omega$  open of  $R^3$ , of border  $\Gamma = \Gamma_1 \cup \Gamma_2$ .

One must solve the equation [eq. 1]  $C_L$  on  $\Omega \times ]0, t[$  with the boundary conditions:

$$\begin{cases} C_L = C_L^d & \text{sur } \Gamma_1 \\ \mathbf{J} \cdot \mathbf{n} = \phi & \text{sur } \Gamma_2 \end{cases} \quad 3$$

with  $\mathbf{J} = -D_L \nabla C_L + \frac{D_L V_H}{RT} C_L \nabla \sigma_H$

and initial conditions  $C_L(t=0)$  (and  $C_T(t=0)$  ).

Either  $v$  a sufficiently regular function cancelling itself on  $\Gamma_1$ , the variational formulation of the problem is written:

$$\int_{\Omega} D^*(C_L) \frac{dC_L}{dt} v d\Omega + \int_{\Omega} D_L \nabla C_L \cdot \nabla v d\Omega - \int_{\Omega} \nabla v \cdot \left( \frac{D_L C_L V_H}{RT} \nabla \sigma_H \right) d\Omega = \int_{\Omega} r_{vol} v d\Omega + \int_{\Gamma_2} \phi v d\Gamma_2 \quad 4$$

with:

$$D^*(C_L) = \frac{C_L + C_T(1 - \theta_T)}{C_L} \quad \phi = \left( D_L \nabla C_L - \frac{D_L V_H}{RT} C_L \nabla \sigma_H \right) \cdot \mathbf{n} \quad \text{and}$$

$$r_{vol} = -\theta_T \frac{dN_T}{d\varepsilon_{eq}^p} \frac{d\varepsilon_{eq}^p}{dt} = 0$$

the numerical resolution by finite elements is thus similar to that of the nonlinear thermal, and leans on  $\theta$ , modulo two characteristics:

- the term formula  $r_{vol}$  is nonlinear, as in the modelization of drying [R7.01.12], and will be integrated explicitly;
- the formula  $\int_{\Omega} \nabla v \cdot \left( \frac{D_L C_L V_H}{RT} \nabla \sigma_H \right) d\Omega$  is asymmetric, and will be also deferred to the second member, as in [1] and will be integrated explicitly.

This explicit discretization of these two terms is not constraining: indeed the resolution of the equation [eq. 4] is carried out with each time step, chained with a mechanical resolution.

On the other hand the formula  $\int_{\Omega} \nabla v \cdot \left( \frac{D_L C_L V_H}{RT} \nabla \sigma_H \right) d\Omega$  requires to apply a variation in temperature  $\nabla T_{ini}$  presumed uniform in the element. The second calculated elementary member is:  $\int_{\Omega} \nabla T_{ini} K \nabla v d\Omega$  where  $K$  is the tensor of thermal conductivities.

Knowing the initial conditions  $C_L^0 = C_L(t=0)$  and  $C_T^0 = C_T(t=0)$ , one calculates  $C_{tot}^0 = C_L^0 + C_T^0$ .

and one adimensionne the unknown (concentration); the variational formulation is then:

$$\int_{\Omega} D^* \frac{dc_L}{dt} v d\Omega + \int_{\Omega} D_L \nabla c_L \cdot \nabla v d\Omega = \int_{\Omega} \nabla v \cdot \left( \frac{D_L V_H}{RT} c_L \nabla \sigma_H \right) d\Omega + \int_{\Omega} \bar{r}_{vol} v d\Omega + \int_{\Gamma_2} \bar{\phi} v d\Gamma_2$$

$$\text{with: } c_L = \frac{C_L}{C_{tot}^0} \quad \bar{r}_{vol} = \frac{-\theta_T}{C_{tot}^0} \frac{dN_T}{d\varepsilon_{eq}^p} \frac{d\varepsilon_{eq}^p}{dt} \quad \text{and } \bar{\phi} = \left( -D_L \nabla c_L + \frac{D_L V_H}{RT} c_L \nabla \sigma_H \right) \cdot \mathbf{n}$$

the coefficient of heat capacity  $D^*$  is a field represented by a command variable (NEUT1).

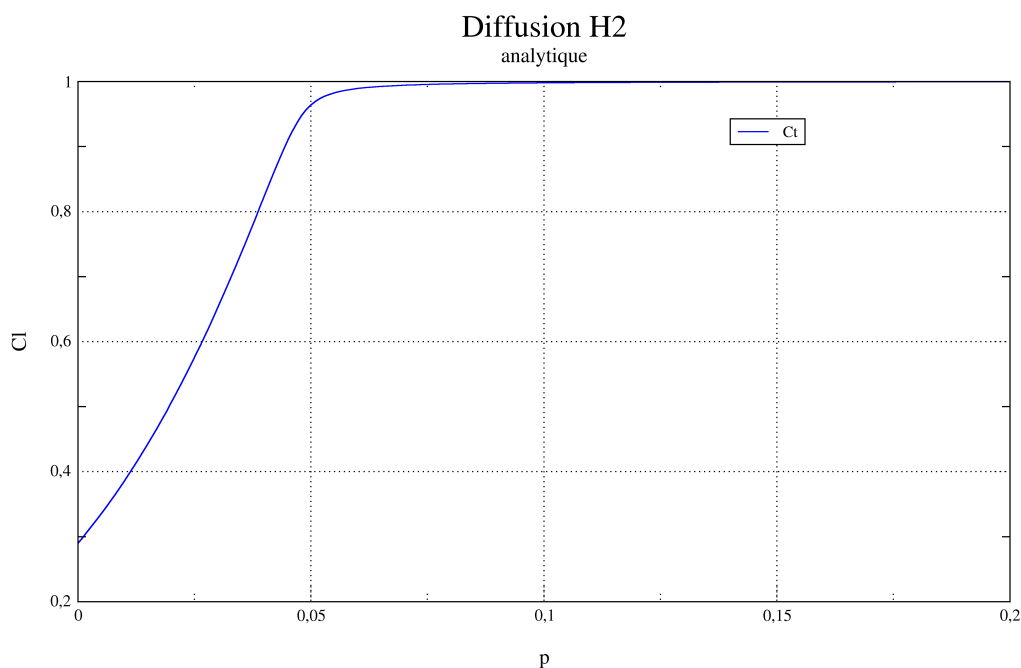
In a volume element isolated ( $\nabla C_L = 0$  on the border) and charged so that the stress state and of strains is uniform, the total concentration  $C_{tot} = C_l + C_t$  is constant.

The analytical solution described in [1] is obtained directly in the form of the quadratic equation:  $C_T^2 - B C_T + N_T C_{tot} = 0$ , with  $B = N_L / K_T + C_{tot} + N_T$ : one of the two solutions would lead to negative values of  $C_L$ , the solution is thus.

$$C_T = 1/2 (B - \sqrt{B^2 - 4 N_T C_{tot}})$$

the representation of  $C_T(\varepsilon_{eq}^p)$   $C_L(\varepsilon_{eq}^p)$  analytically and numerically in [1] is:

## 2.2 Results of reference



## 2.3 Uncertainty on the solution

Uncertainty lower than 1 % .

## 2.4 Bibliographical references

- 1) "Hydrogen near transport has blunting ace tip" A.H.M. Krom, R.W.J.Koers, A.Bakker in "Newspaper of the Mechanics and Physics of Solids" 45 (1999) 971-992 Modelization

## 3 A Characteristic

### 3.1 of the modelization A Mechanical

: modelization C\_PLAN Thermal  
: modelization PLAN\_DIAG Characteristics

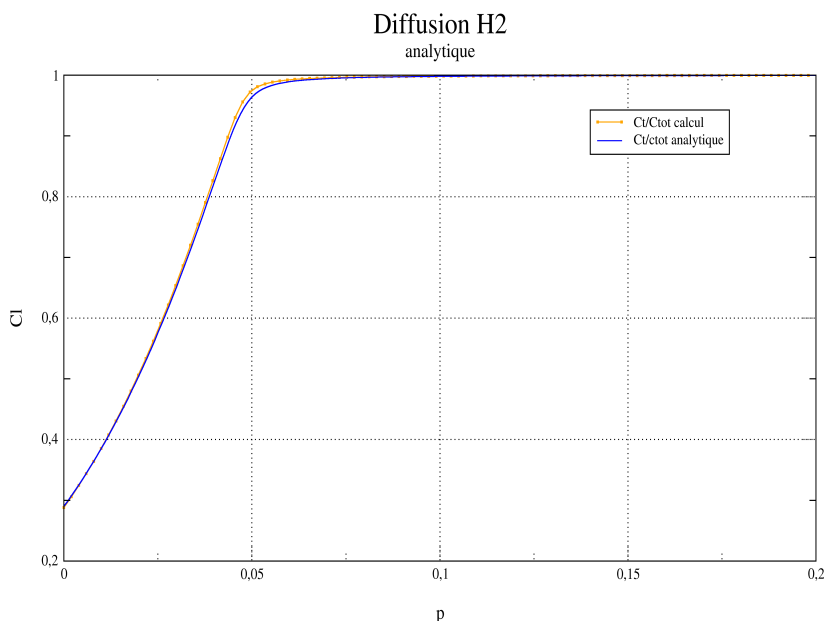
### 3.2 of the mesh an element

QUAD4. Many nodes: 4 Quantities

### 3.3 tested and results the analytical

solution being obtained according to the plastic strain, one tests initially this value according to time, then the hydrogen concentration according to time: Identification

Time Reference	Tolerance	% P1 11000000.0
0.0198 0,10% P1		0.10%
0.0297 0,10% P1		0.10%
0.0396 0,10% P1		0.10%
0.0495 0,10% P1		0.10%
0.0594 0,10% P1		0.10%
0.100980 0,10% P1		0.10%
0.1980 0,10% CT2		0.10%
0.5058 0,10% CT2		0.10%
0.6522 0.3% CT2		12000000.0
0.8239 0.4% CT2		12500000.0
0.964 1.0% CT2		13000000.0
0.989 0.4% CT2		15100000.0
0.99807 0,10% CT2		0.10%
0.999628 0,10% Modelization		0.10%



Warning : The  
provided as a convenience.

in part and is

## 4 B Characteristic

### 4.1 of the modelization B Mechanical

: modelization 3D Thermal  
: modelization 3D\_DIAG Characteristics

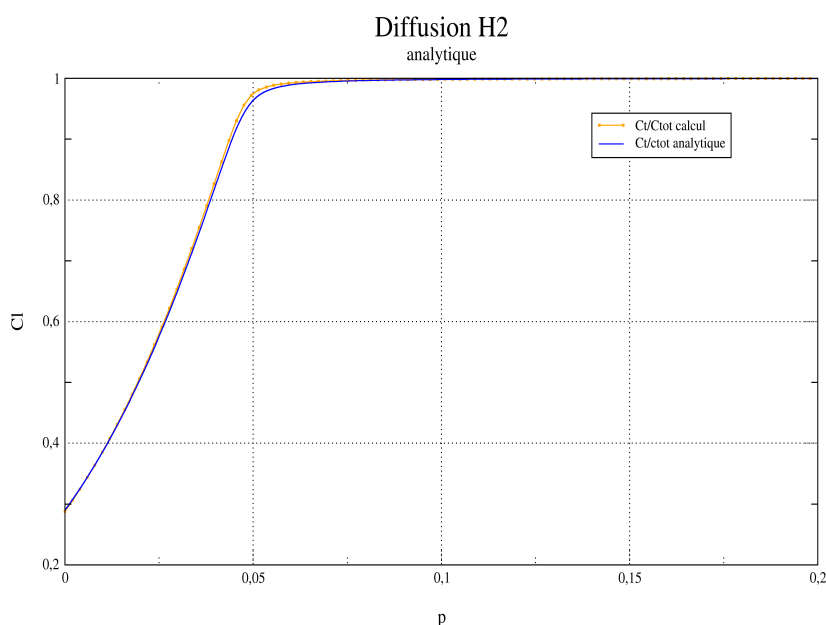
### 4.2 of the mesh an element

HEXA8. Many nodes: 8 Quantities

### 4.3 tested and results the analytical

solution being obtained according to the plastic strain, one tests initially this value according to time, then the hydrogen concentration according to time: Identification

Time Reference	Tolerance	% P1 11000000.0
0.0198 0,10% P1		0.10%
0.0297 0,10% P1		0.10%
0.0396 0,10% P1		0.10%
0.0495 0,10% P1		0.10%
0.0594 0,10% P1		0.10%
0.100980 0,10% P1		0.10%
0.1980 0,10% CT2		0.10%
0.5058 0,10% CT2		0.10%
0.6522 0.3% CT2		12000000.0
0.8239 0.4% CT2		12500000.0
0.964 1.0% CT2		13000000.0
0.989 0.4% CT2		15100000.0
0.99807 0,10%		0.10%
0.999628 0,10% Summary		0.10%





## 5 of the results the calculated

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solution is very close to the analytical and valid solution the methodology of simulation of the hydrogen diffusion in a steel.